



# ***Synchrotron X-ray diffraction Studies of Ceramic Oxides***

**J. Bashir**

Pakistan Institute of Nuclear Science & Technology  
P.O. Nilore, Islamabad

## ***Synchrotron X-ray diffraction Studies of Ceramic Oxides***

Recent years have seen a dramatic growth in the applications of synchrotron powder diffraction measurements. Powder diffraction beam lines at synchrotron user facilities offer advantages of intensity, resolution and tunability, relative to conventional X-ray generators. SR powder diffraction has been used for a variety of experiments where high accuracy of data is important in crystal structure determination including both nuclear and magnetic crystal structure, space group resolution, charge distribution, and phase transformations.

## ..... Bit of History

- 1947 First synchrotron radiation seen
- 1961 First SR facility at NBS, Washington, USA
  - 180MeV
  - Measurement of Absorption Spectra of rare gases
- 1979 First experiment to assess the advantages of SR diffraction over conventional diffraction was carried out at Laboratory for High Energy Physics (KEK), Japan.

By 1983, foundations of SR powder diffraction were well laid out. Since that time the techniques employing SR are now having a major impact on several areas of physics, chemistry and biological sciences. It has now become possible to perform entirely new type of experiments, for example, high temperature, high pressure, magnetic scattering and kinetic or time – resolved crystallography.

# Characteristics of Radiation

	Electrons			X-Rays			Hard X-rays		Neutrons		
Energy	100keV	200keV	500keV	Fe 7keV	Cu9keV	Mo 20keV	35keV	80keV	1 meV	10 meV	100 meV
Wavelength (nm)	0.0037	0.0025	0.0014	0.194	0.154	0.071	0.035	0.015	0.9	0.29	0.09
Velocity (m/s)	$1.7 \times 10^8$	$2.1 \times 10^8$	$2.6 \times 10^8$	$3 \times 10^8$	$3 \times 10^8$	$3 \times 10^8$	$3 \times 10^8$	$3 \times 10^8$	430	1400	3100
Temperature (K)	$1 \times 10^9$	$2 \times 10^9$	$5 \times 10^9$	$0.8 \times 10^8$	$1 \times 10^8$	$2 \times 10^8$	$3.5 \times 10^8$	$8 \times 10^8$	12	116	580
Penetration in Fe	~100nm			~10 $\mu$ m			1mm-1cm		~ 5 cm		

# Synchrotron Techniques

## Main techniques

- Absorption based: characteristic absorption of X-rays by samples, especially in the region of absorption edges used to deduce local structure
- Attenuation/absorption used in radiography/tomography
- Diffraction uses a monochromatic beam for structure determination, stress, phase identification, etc

## **The diverse uses of synchrotrons**

- **Medical imaging and therapy**
- **Environment**
- **Forensics**
- **Manufacturing**
- **Medicine and pharmaceuticals**
- **Agriculture**
- **Minerals**
- **Micromachining**
- **Materials sciences and engineering**

# Oxide based materials

- The majority of advanced materials used in magnetic, conductivity, superconductivity, ferroelectric, catalytic and battery applications are solid metal oxides. Metal oxide chemistry is dominated by classes of materials having crystal structures derived from simpler parent structures such as perovskite or rutile.
- Small lattice distortions, which are critical to the key electronic and physical properties of these oxides, usually lead to lower symmetries and superstructures. These distortions are characterised by subtle peak splittings and the appearance of weak superlattice reflections in diffraction data.
- The detection and understanding of such distortions requires the high resolution afforded by synchrotron radiation



# Science

The impact of powder diffraction and the use of the Rietveld method for structural crystallography is

**MASSIVE**

In the last decade, powder diffraction has been the technique of choice to provide vital structural insight in diverse areas:



# Science

- **High Temperature Cuprate and other Oxide Superconductors:**

- Structure and crystal-chemistry of the high-Tc superconductor  $\text{YBa}_2\text{Cu}_3\text{O}_{7-x}$ , *Nature*, **327**, 310-312 (1987)
- Superconductivity near 30-K without copper - the  $\text{Ba}_{0.6}\text{K}_{0.4}\text{BiO}_3$  Perovskite, *Nature*, **332**, 814-816 (1988)
- Synthesis and superconducting properties of the strontium copper oxy-fluoride  $\text{Sr}_2\text{CuO}_2\text{F}_{2+d}$ , *Nature*, **369**, 382(1994)
- Cation effects in doped  $\text{La}_2\text{CuO}_4$  superconductors, *Nature*, **394**, 157-159 (1998)
- Systematic cation disorder effects in  $\text{L}_{1.85}\text{M}_{0.15}\text{CuO}_4$  superconductors, *Phys. Rev. Lett.*, **83**, 3289-3292 (1999)

- **MgB<sub>2</sub> and Borocarbide Superconductors:**

- Structure of the 13-K superconductor  $\text{La}_3\text{Ni}_2\text{B}_2\text{N}_3$  and the related phase  $\text{LaNiBN}$ ", *Nature*, **372**, 759-761 (1994)
- $\text{MgB}_2$  superconducting thin films with a transition temperature of 39 Kelvin", *Science*, **292**, 1521-1523 (2001)
- Superconductivity at 39 K in magnesium diboride", *Nature*, **410**, 63(2001)

# Science

- **C<sub>60</sub> and its Superconducting Derivatives:**
  - Superconductivity at 28 K in Rb<sub>x</sub>C<sub>60</sub>, *Phys. Rev. Lett.*, **66**, 2830(1991)
  - Intercalation of ammonia into K<sub>3</sub>C<sub>60</sub>, *Nature*, **364**, 425-427 (1993)
  - Crystal-structure, bonding, and phase-transition of the superconducting Na<sub>2</sub>CsC<sub>60</sub> Fulleride, *Science*, **263**, 950-954 (1994)
  - Structural and electronic properties of the noncubic superconducting fullerenes A<sub>4</sub>C<sub>60</sub> (A = Ba, Sr), *Phys. Rev. Lett.*, **83**, 2258(1999)
- **Cathode and Electrolytic Materials for Portable, Rechargeable Batteries:**
  - Crystal-structure of the polymer electrolyte poly(ethylene Oxide)<sub>3</sub>:LiCF<sub>3</sub>SO<sub>3</sub>, *Science*, **262**, 883-885 (1993)
  - Synthesis of layered LiMnO<sub>2</sub> as an electrode for rechargeable lithium batteries, *Nature*, **381**, 499-500 (1996)
  - Structure of the polymer electrolyte poly(ethylene oxide)<sub>6</sub>:LiAsF<sub>6</sub>, *Nature*, **398**, 792-794 (1999)
  - Ionic conductivity in crystalline polymer electrolytes, *Nature*, **412**, (2001)

# Science

## ● Giant Magneto-Resistive Materials

- Simultaneous Structural, Magnetic, and Electronic-Transitions in  $\text{La}_{1-x}\text{Ca}_x\text{MnO}_3$  with  $x=0.25$  and  $0.50$ , *Phys. Rev. Lett.*, **75**, 4488-4491 (1995)
- Colossal magnetoresistance without  $\text{Mn}^{3+}/\text{Mn}^{4+}$  double exchange in the stoichiometric pyrochlore  $\text{Tl}_2\text{Mn}_2\text{O}_7$ , *Science*, **273**, 81-84 (1996)
- Lattice effects and magnetic order in the canted ferromagnetic insulator  $\text{La}_{0.875}\text{Sr}_{0.125}\text{MnO}_{3+d}$ ", *Phys. Rev. Lett.*, **76**, 3826-3829 (1996)
- Direct observation of lattice polaron formation in the local structure of  $\text{La}_{1-x}\text{Ca}_x\text{MnO}_3$ ", *Phys. Rev. Lett.*, **77**, 715-718 (1996)
- Colossal magnetoresistance in Cr-based chalcogenide spinels", *Nature*, **386**, 156-159 (1997)
- Electrostatically driven charge-ordering in  $\text{Fe}_2\text{OBO}_3$ , *Nature*, **396**, 655-658 (1998)
- Optimal T-C in layered manganites: Different roles of coherent and incoherent lattice distortions, *Phys. Rev. Lett.*, **83**, 1223-1226 (1999)
- Formation of isomorphous  $\text{Ir}^{3+}$  and  $\text{Ir}^{4+}$  octamer and spin dimerisation in the spinel  $\text{CuIr}_2\text{S}_4$ , *Nature*, **416** 155-158 (2002)

# Science

- **First Metal Oxide Hydride:**

- The hydride anion in an extended transition metal oxide array:  $\text{LaSrCoO}_3\text{H}_{0.7}$ , *Science*, **295**, 1882 (2002)

- **New Dielectric Materials:**

- Enhancement of the dielectric-constant of  $\text{Ta}_2\text{O}_5$  through substitution with  $\text{TiO}_2$ , *Nature*, **377**, 215-217 (1995)

- **Highly-Reactive Molecular Species:**

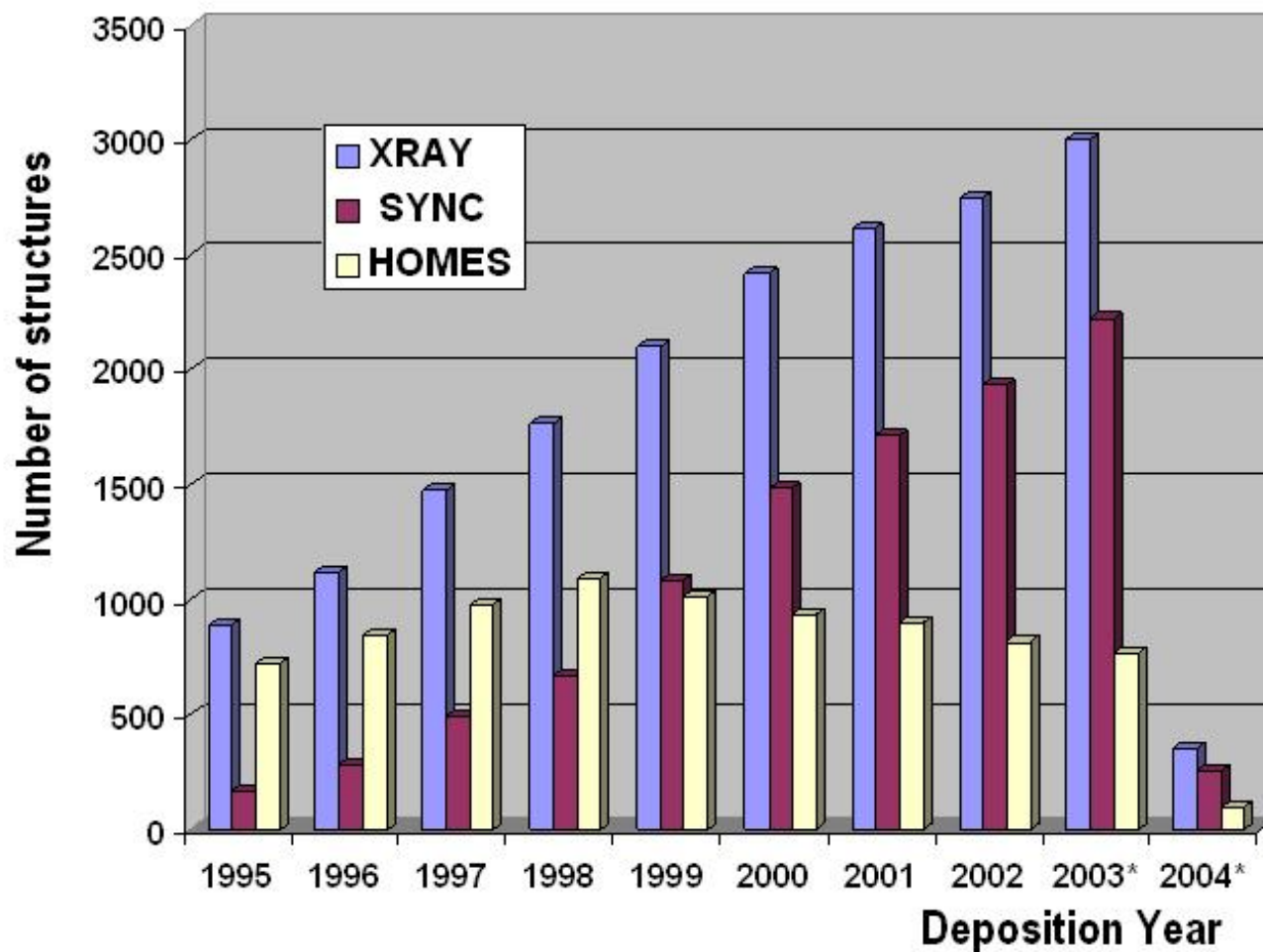
- Crystal and molecular-structures of rhenium heptafluoride, *Science*, **263**, 1265-1267 (1994)

- **Magnetic Nanomaterials:**

- Monodisperse FePt nanoparticles and ferromagnetic FePt nanocrystal superlattices, *Science*, **287**, 1989-1992 (2000)
- Size-dependent grain-growth kinetics observed in nanocrystalline Fe", *Phys. Rev. Lett.*, **86**, 842-845 (2001)

Fig.1

All X-ray, Synchrotron vs Home Sources, As of May 5, 2004



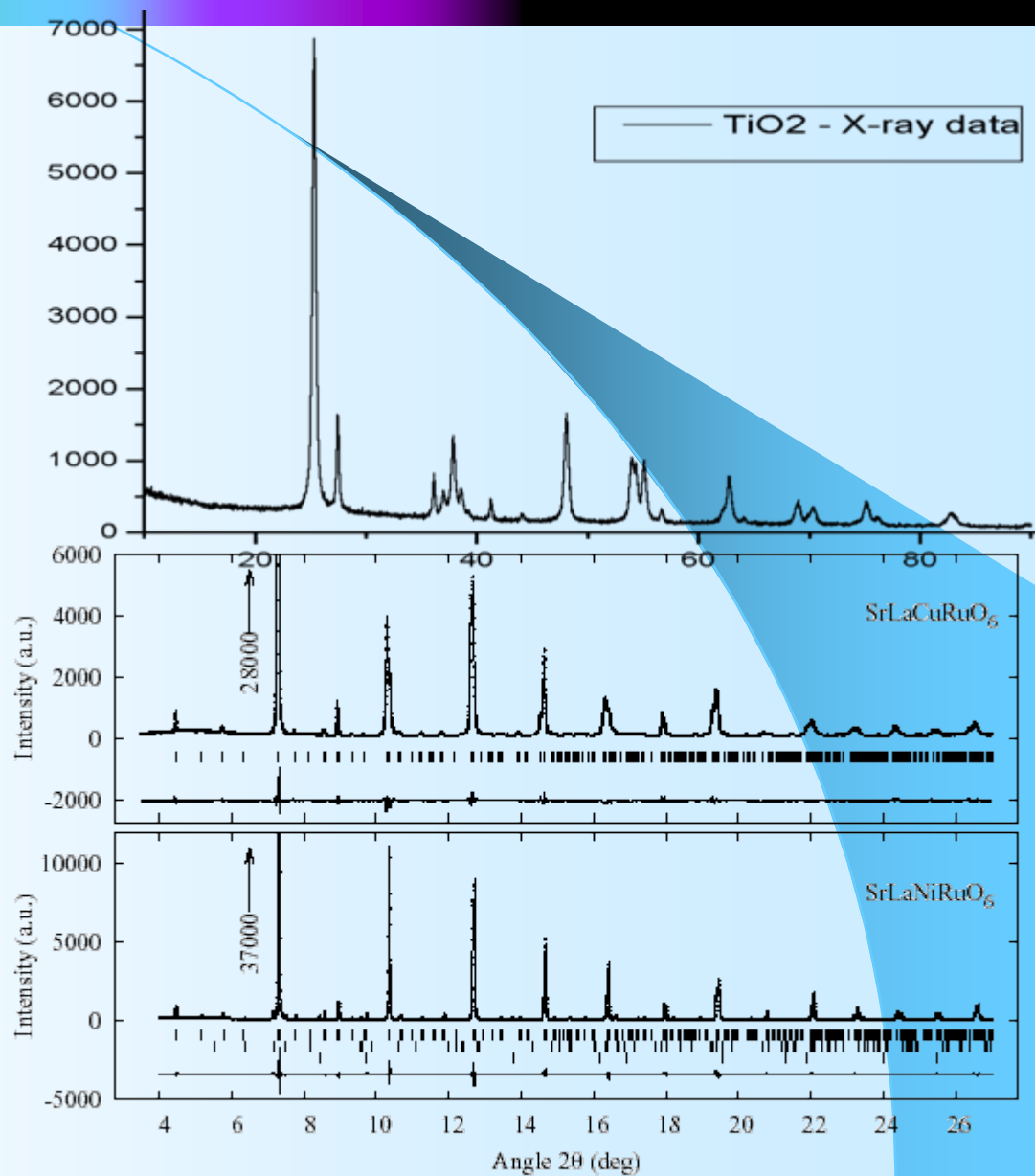
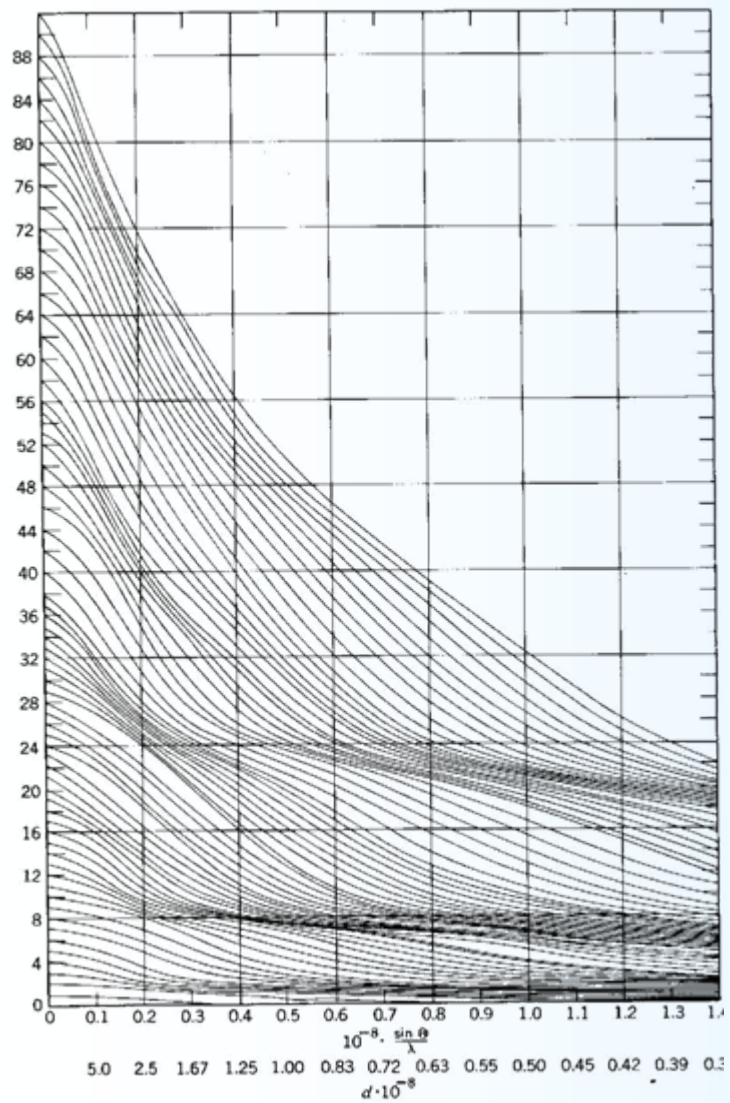
# Properties of synchrotron light

- **High brightness:** synchrotron light is extremely intense ( $10^{12}$  times more intense than that from conventional x-ray tubes) and are highly collimated.
- **Wide energy spectrum:** synchrotron light is emitted with energies ranging from infrared light to hard, energetic (short wavelength) x-rays.
- **Tunable:** through sophisticated monochromators and insertion devices it is possible to obtain an intense beam of any selected wavelength.



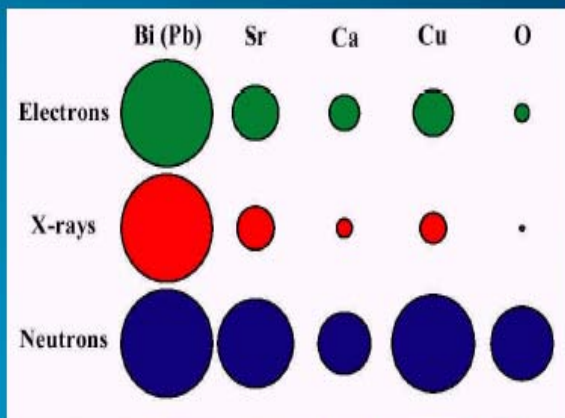
## Higher Flux

- Smaller Samples (milligram)
- Collect more data in a shorter amount of time
- Collect better quality data (increased resolution)
- Tunable energy
  - Spectrum of x-ray energies available for specialized experiments

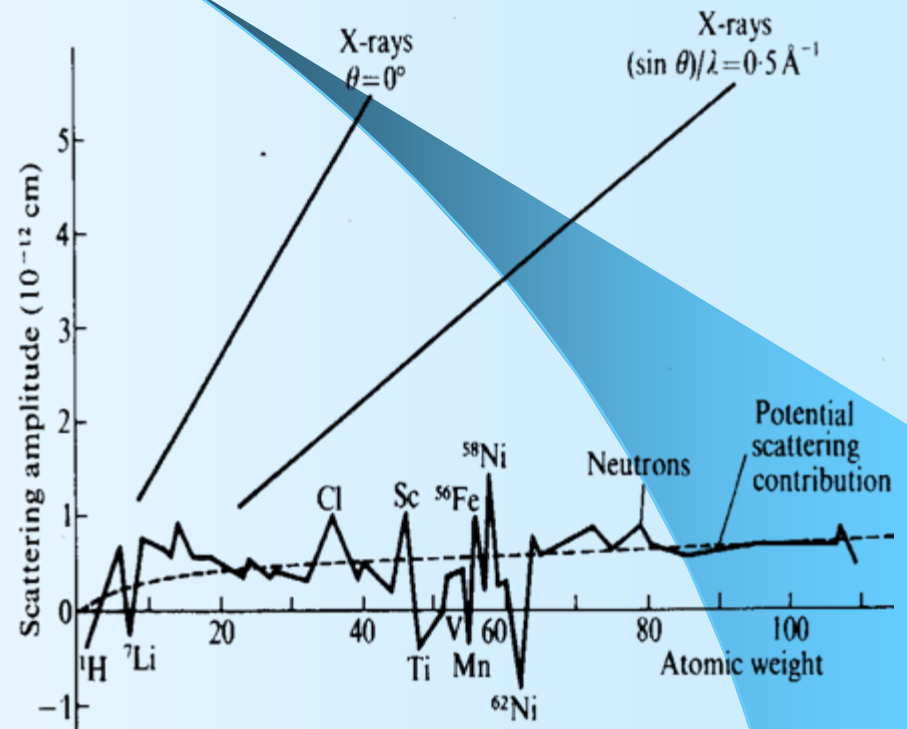




## Relative Scattering Powers of the Elements



Neutrons scatter strongly from light elements  
(Because neutron scattering is a nuclear interaction)

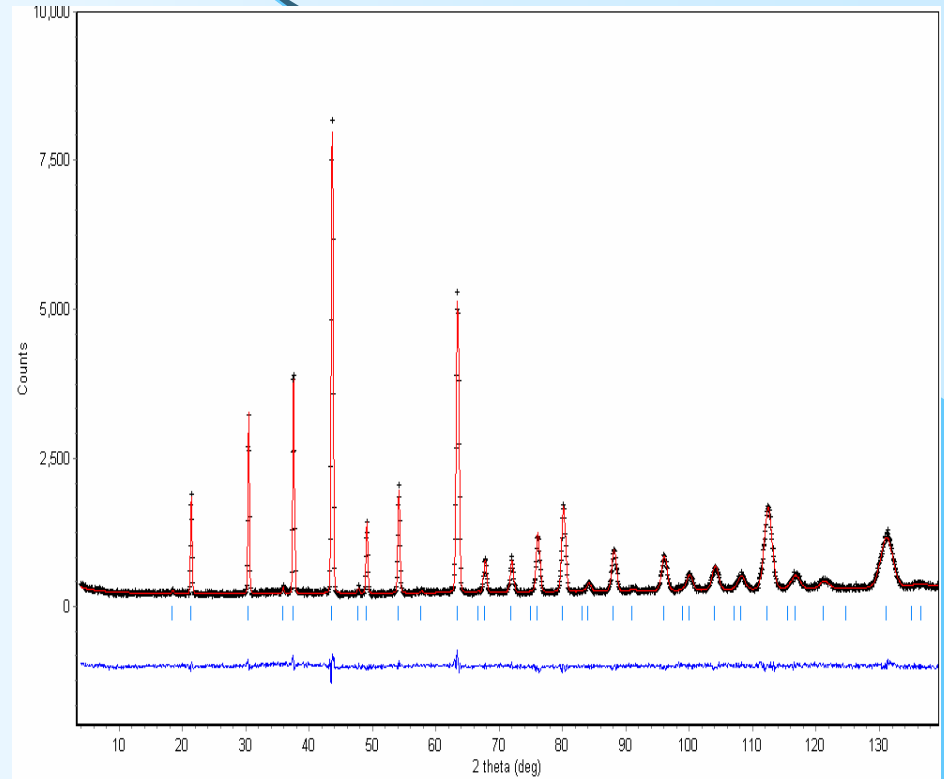


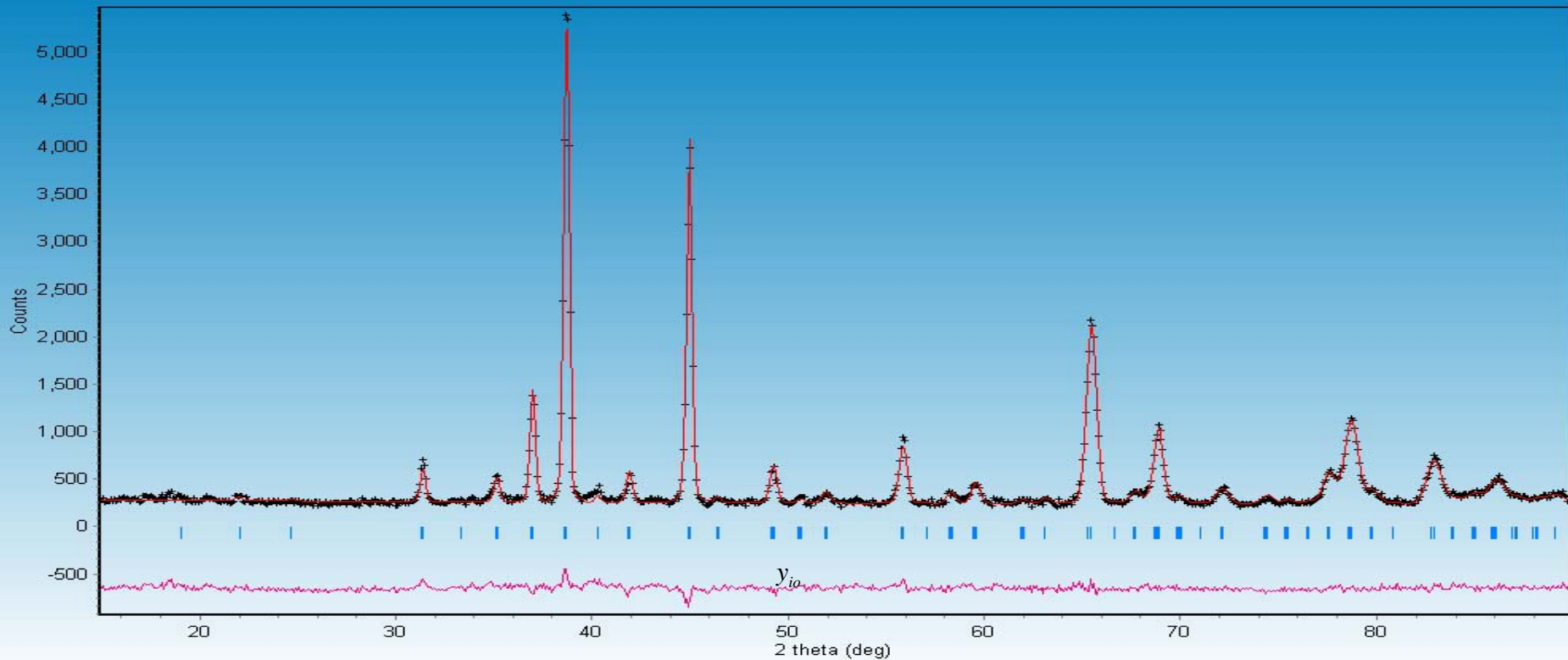


# Rietveld Refinement

# What we start with

- Space Group
- Unit Cell parameters
- Atomic Positions
- Atomic Occupancies





The basis of the Rietveld method is the equation  $y_{ic} = y_{ib} + \sum_p \sum_k G_{ik}^p I_k$  where  $y_{ic}$  is the net intensity calculated at point  $i$  in the pattern,  $y_{ib}$  is the background intensity,  $G_{ik}$  is a normalised peak profile function, and  $I_k = S m_k L_k |F_k|^2 P_k A_k E_k$  is the intensity of the  $k^{\text{th}}$  Bragg reflection.

$$R_p = \frac{\sum |y_{ic} - y_{io}|}{\sum y_{io}} \quad R_p = \left[ \frac{\sum w_i (y_{io} - y_{ic})^2}{\sum w_i y_{io}^2} \right]^{1/2}$$

# Examples



# La<sub>2</sub>NiRuO<sub>6</sub>

- Galasso and Darby\* reported the La<sub>2</sub>NiRuO<sub>6</sub> to be cubic [a = 7.90 Å Space group = Fm3m]
- Cubic with P4<sub>2</sub>32 space group.

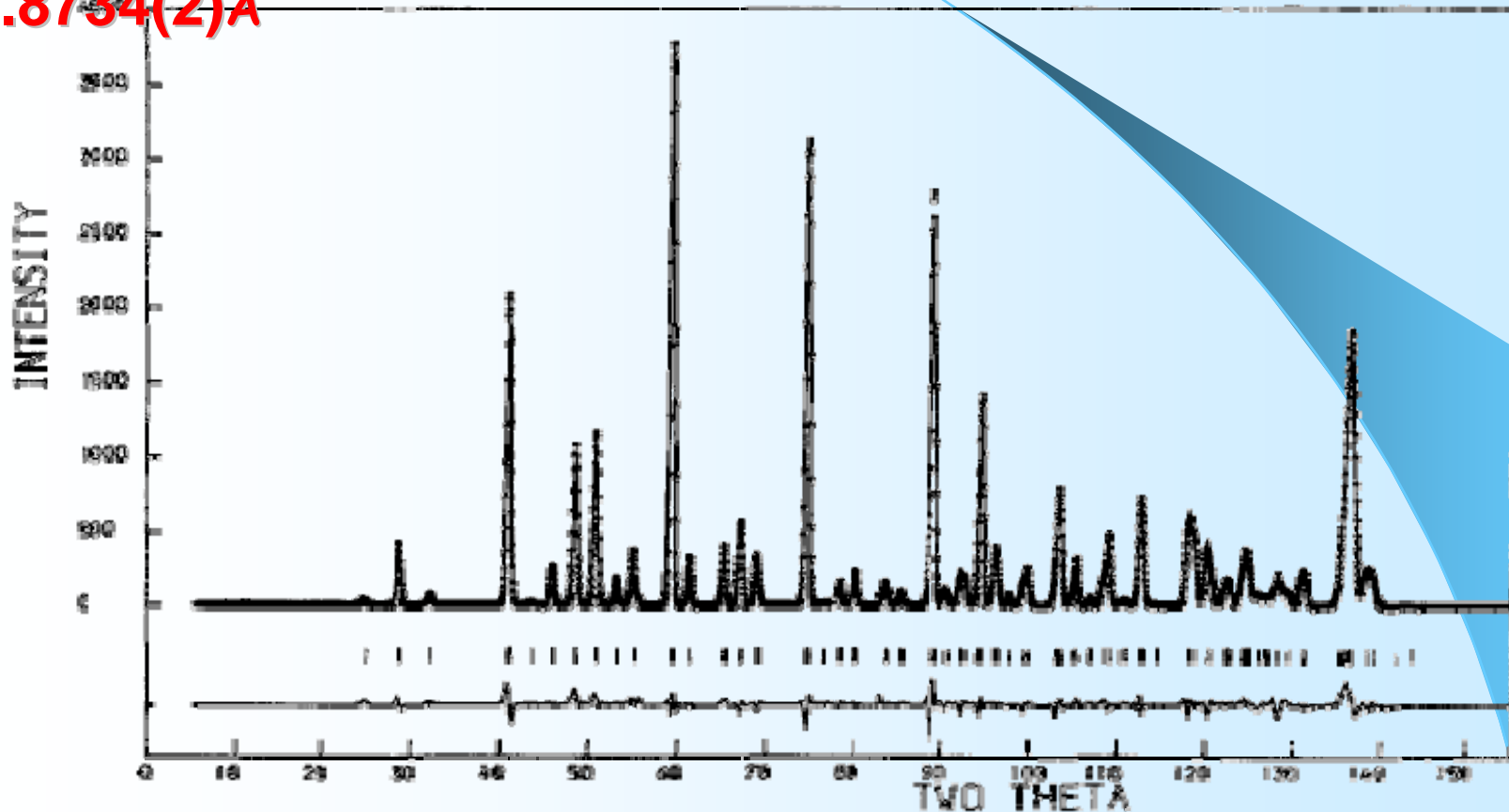
\* F.S. Galasso, *Structure, Properties and Preparation of perovskite-type compounds*.

Pergamon Press Inc. 1969

Fernandez, *et al J. Sol. State Chem.* 32 (1980) 97–104.

# La<sub>2</sub>NiRuO<sub>6</sub> – Neutron Diffraction

Space group Pbnm,  $a = 5.5675(1)\text{\AA}$ ,  $b = 5.5952(1)\text{\AA}$ ,  $c = 7.8734(2)\text{\AA}$

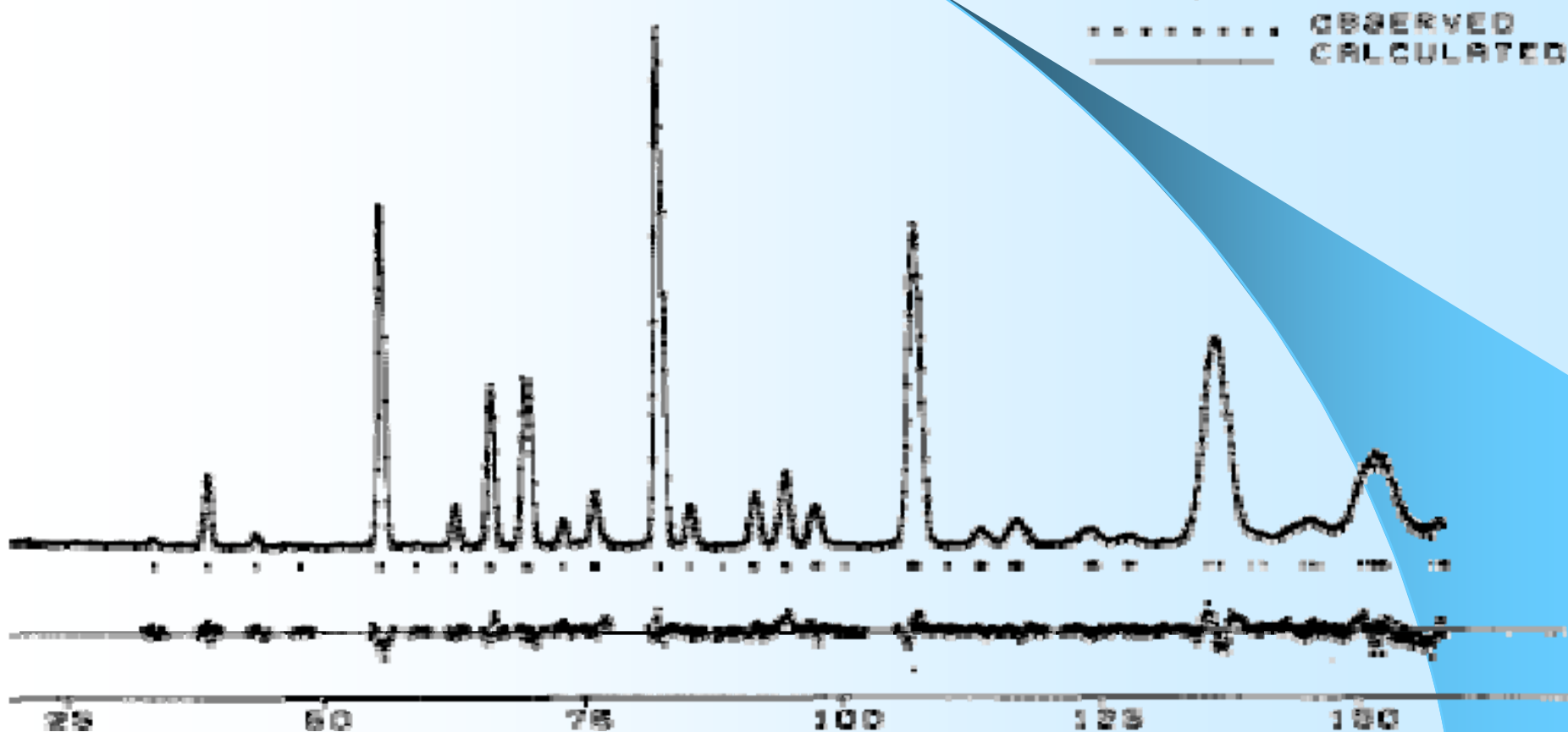


Reference: Battle & Jones, *Mat. Res. Bull.*, 22(1987)1623

# La<sub>2</sub>NiRuO<sub>6</sub> – Neutron Diffraction

Monoclinic, P2<sub>1</sub>/n.

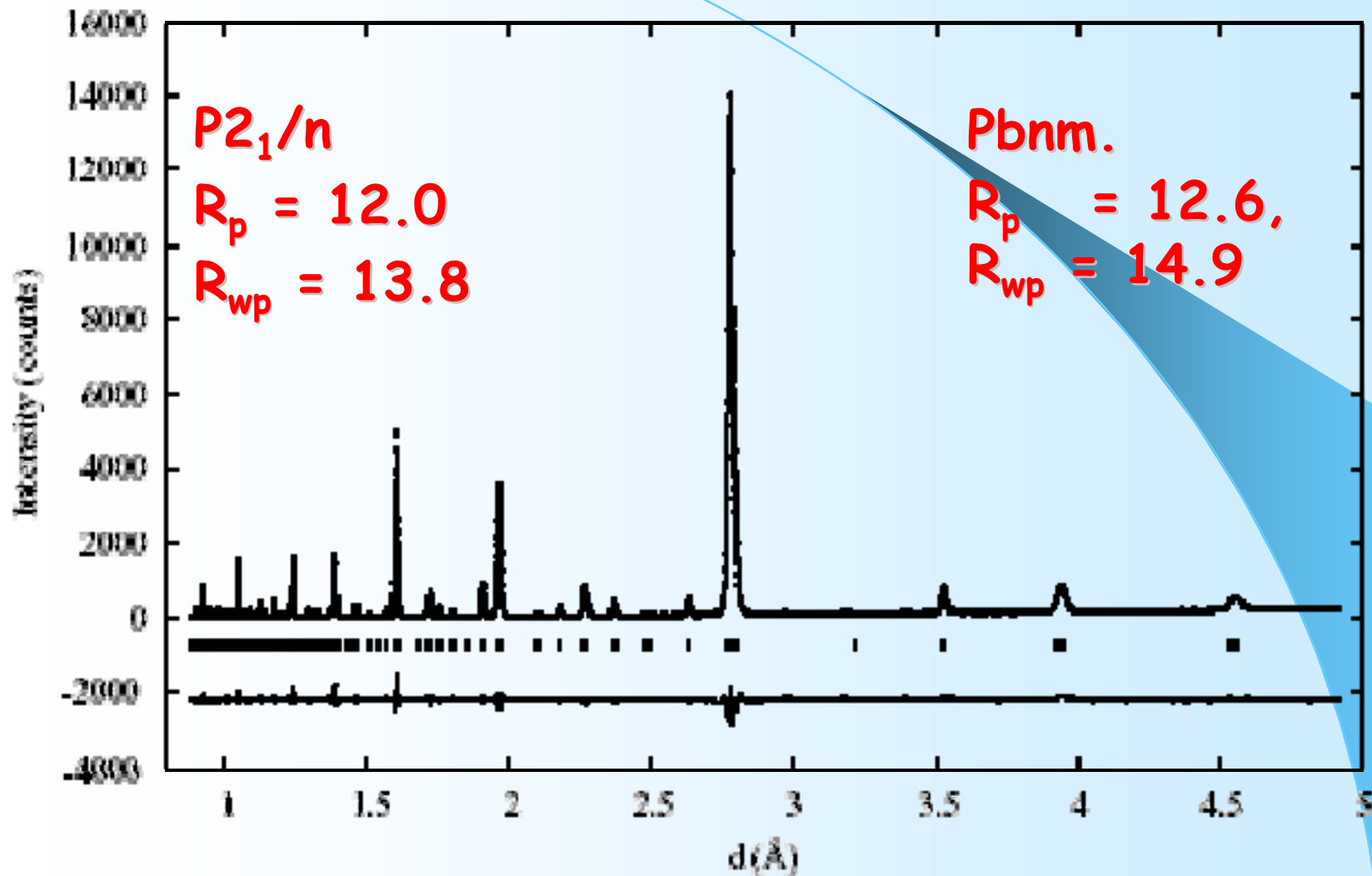
$a = 5.5688(4)\text{\AA}$ ,  $b = 5.5984(4)\text{\AA}$ ,  $c = 7.8764(6)\text{\AA}$ ,  $\beta = 90.18(1)^\circ$



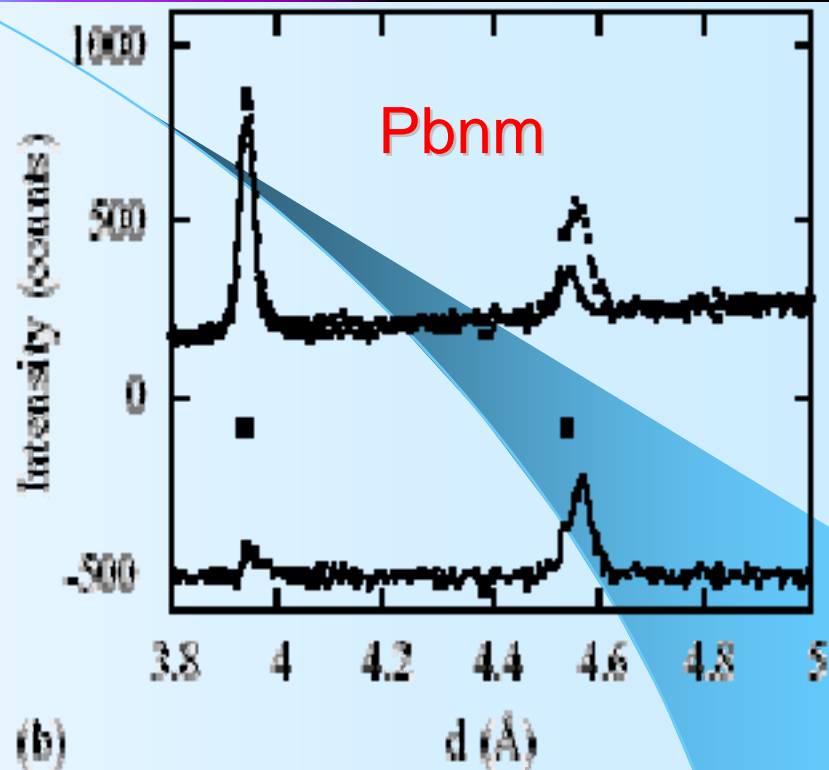
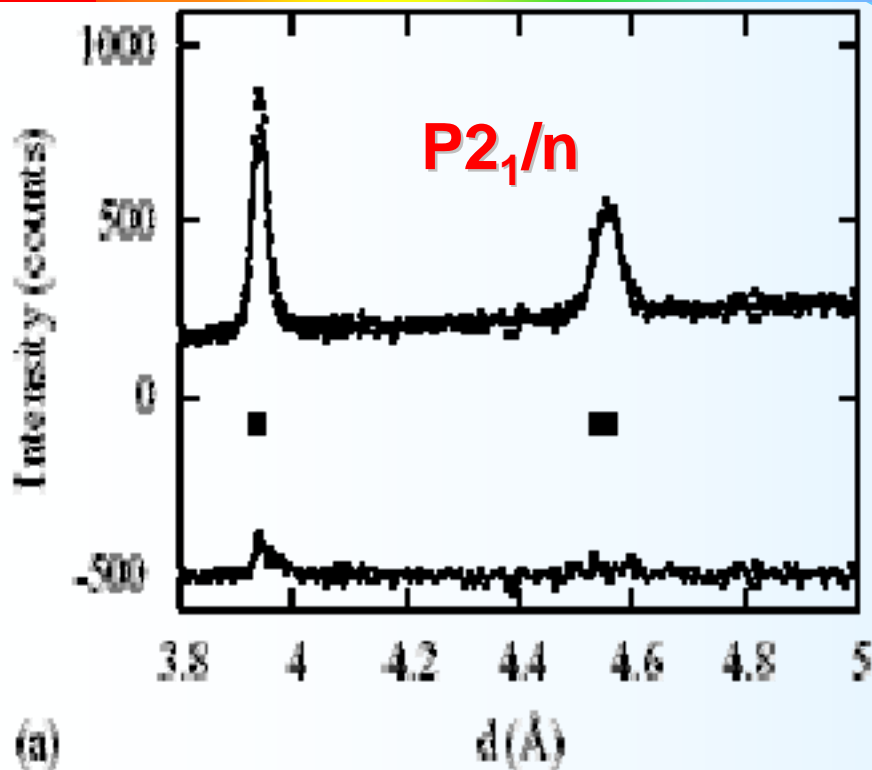
Reference: Seinen et al, Mat. Res. Bull., 22(1987)535.



# La<sub>2</sub>NiRuO<sub>6</sub> – Synchrotron X-ray Diffraction

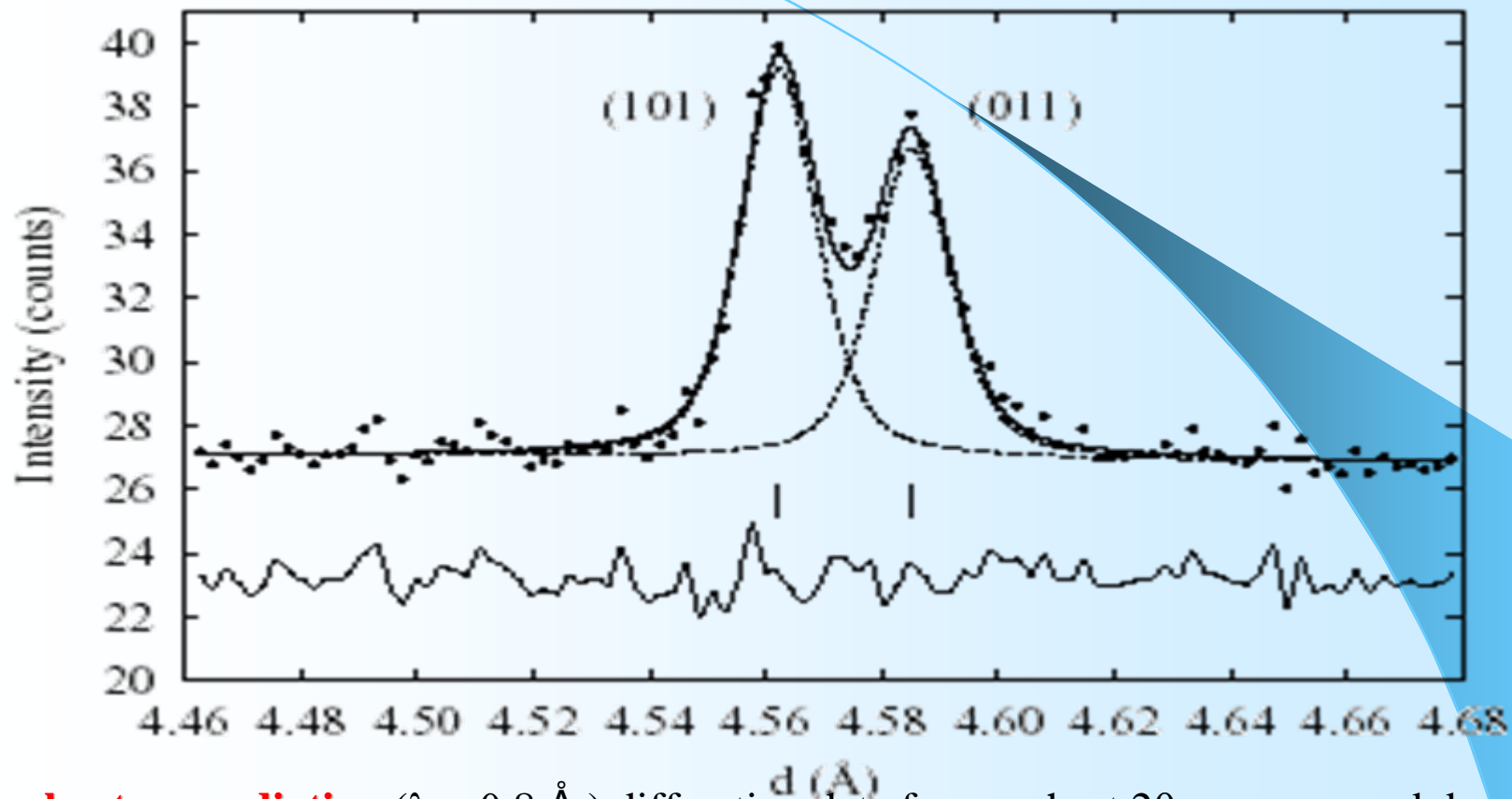


Reference : M. Gateshki et al, *Materials Research Bulletin* 38 (2003) 1661–1668



A section of the **laboratory X-ray** diffraction pattern refined with the  $P2_1/n$  (a) and  $Pbnm$  (b) space groups. The structural model with  $P2_1/n$  space group fits better the experimental data. The Bragg positions calculated for both,  $K\alpha_1$  and  $K\alpha_2$ , are shown.

*Reference : M. Gateshki et al, Materials Research Bulletin 38 (2003) 1661–1668*



**Synchrotron radiation** ( $\lambda = 0.8 \text{ \AA}$ ) diffraction data from a short  $2\theta$  range around  $d = 4.56 \text{ \AA}$ . Lines represent the profile fit obtained with a two-peak model. Data indexed with  $P2_1/n$  space group. The presence of the (0 1 1) reflection in the diffraction pattern confirms the ordered arrangement of the B-cations in  $\text{La}_2\text{NiRuO}_6$

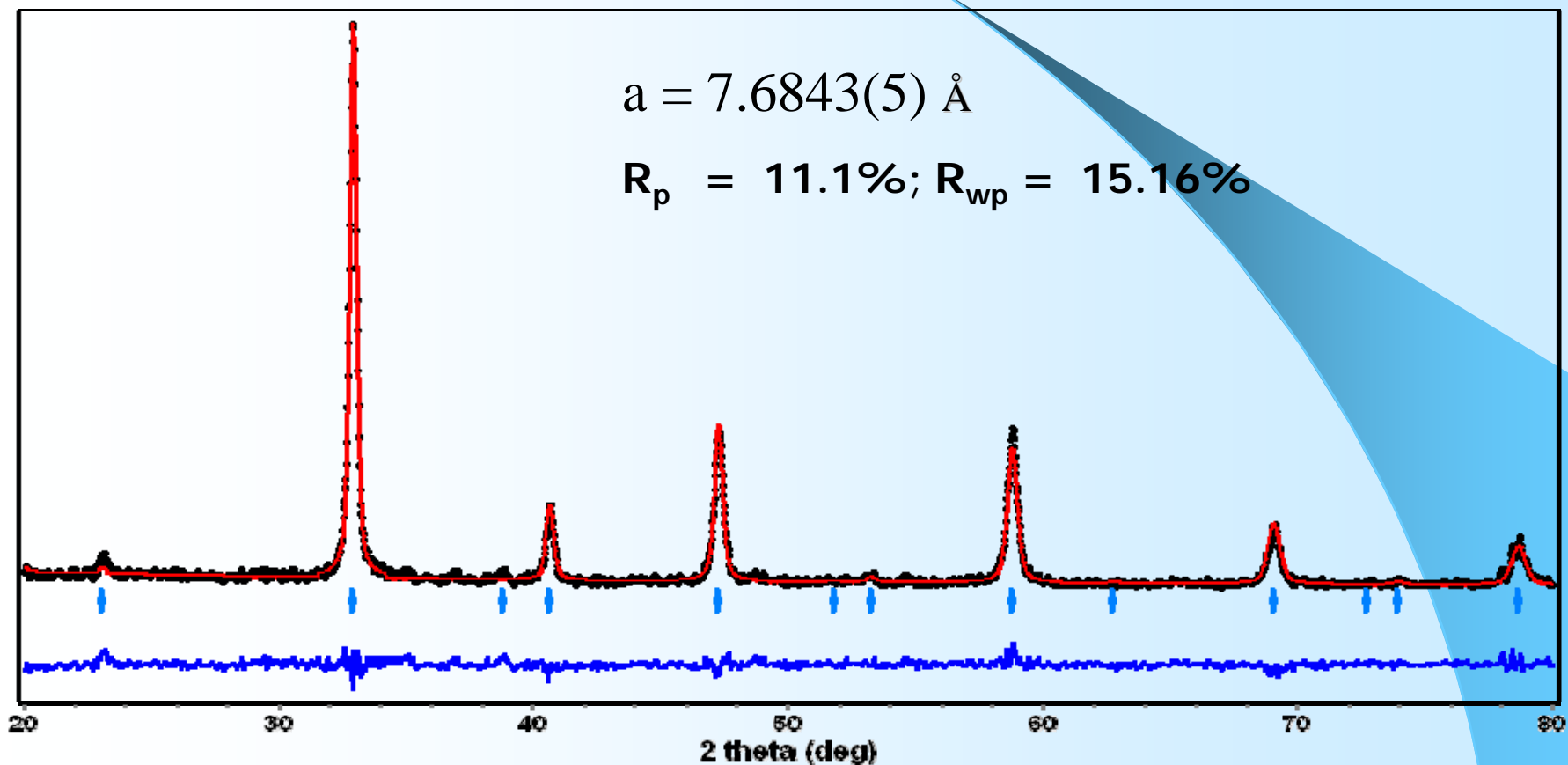
**Reference :** *M. Gateshki et al, Materials Research Bulletin 38 (2003) 1661–1668*

# CaLaMnFeO<sub>6</sub> - Synthesis

Starting Materials: CaCO<sub>3</sub>, La<sub>2</sub>O<sub>3</sub>, MnO<sub>2</sub>, Fe<sub>2</sub>O<sub>3</sub>

- CALCINATION: 1000 C FOR 18 HOURS IN AIR  
1100 C FOR 18 HOURS IN AIR  
1200 C FOR 18 HOURS IN AIR
- PELLETIZATION: 12mm DIAMETER & 2mm THICK PELLETS
- SINTERING: 1250 C FOR 72 HOURS IN AIR
- ANNEALING: 900 C FOR 48 HOURS IN O<sub>2</sub>
- COOLING: FURNACE COOLED

# CaLaMnFeO<sub>6</sub> - Fm3m



The observed (+) and calculated X-ray diffraction pattern of CaLaMnFeO<sub>6</sub> refined in space group Fm3m. Arrows indicate the weak unindexed reflections.

# CaLaMnFeO<sub>6</sub>

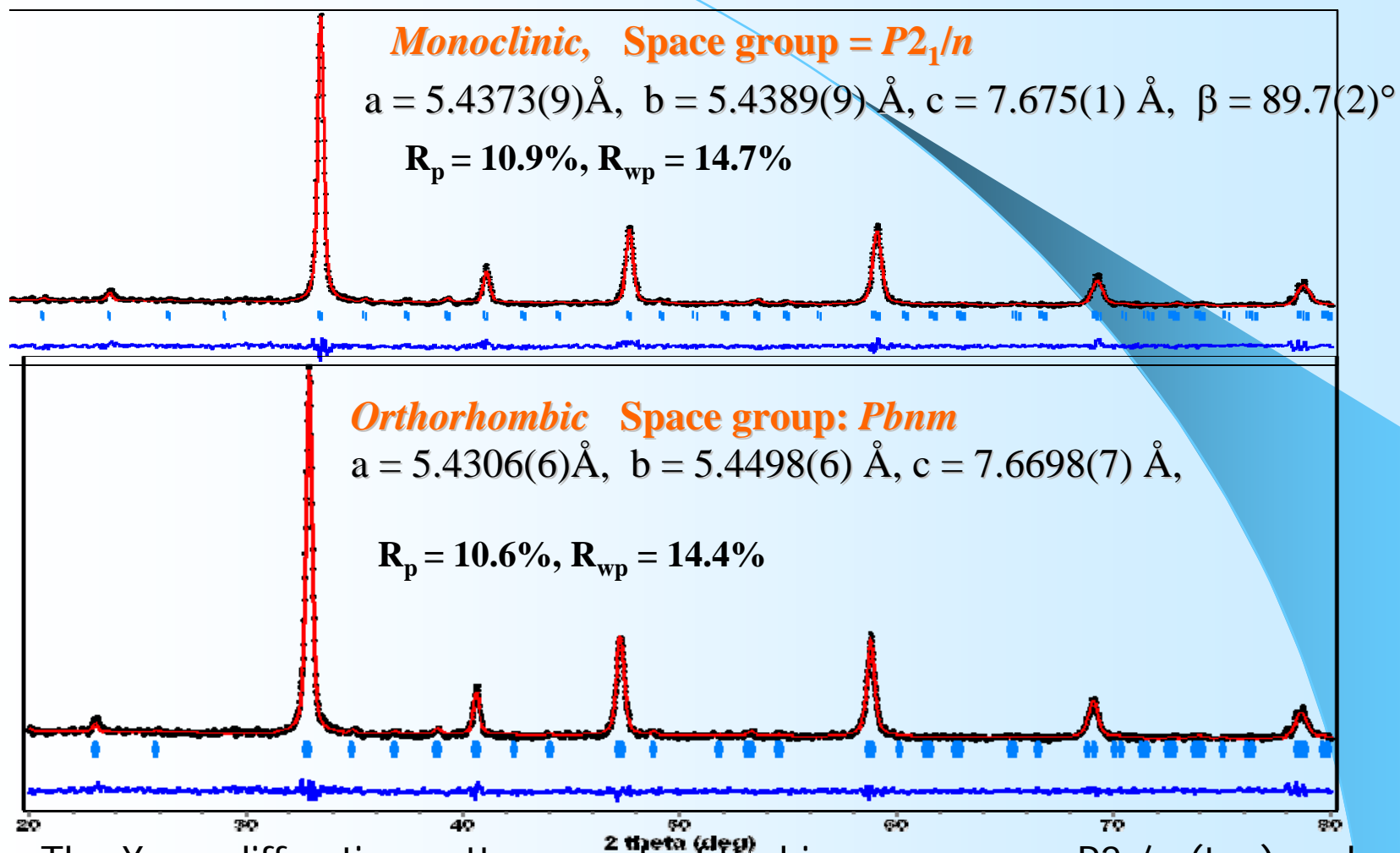
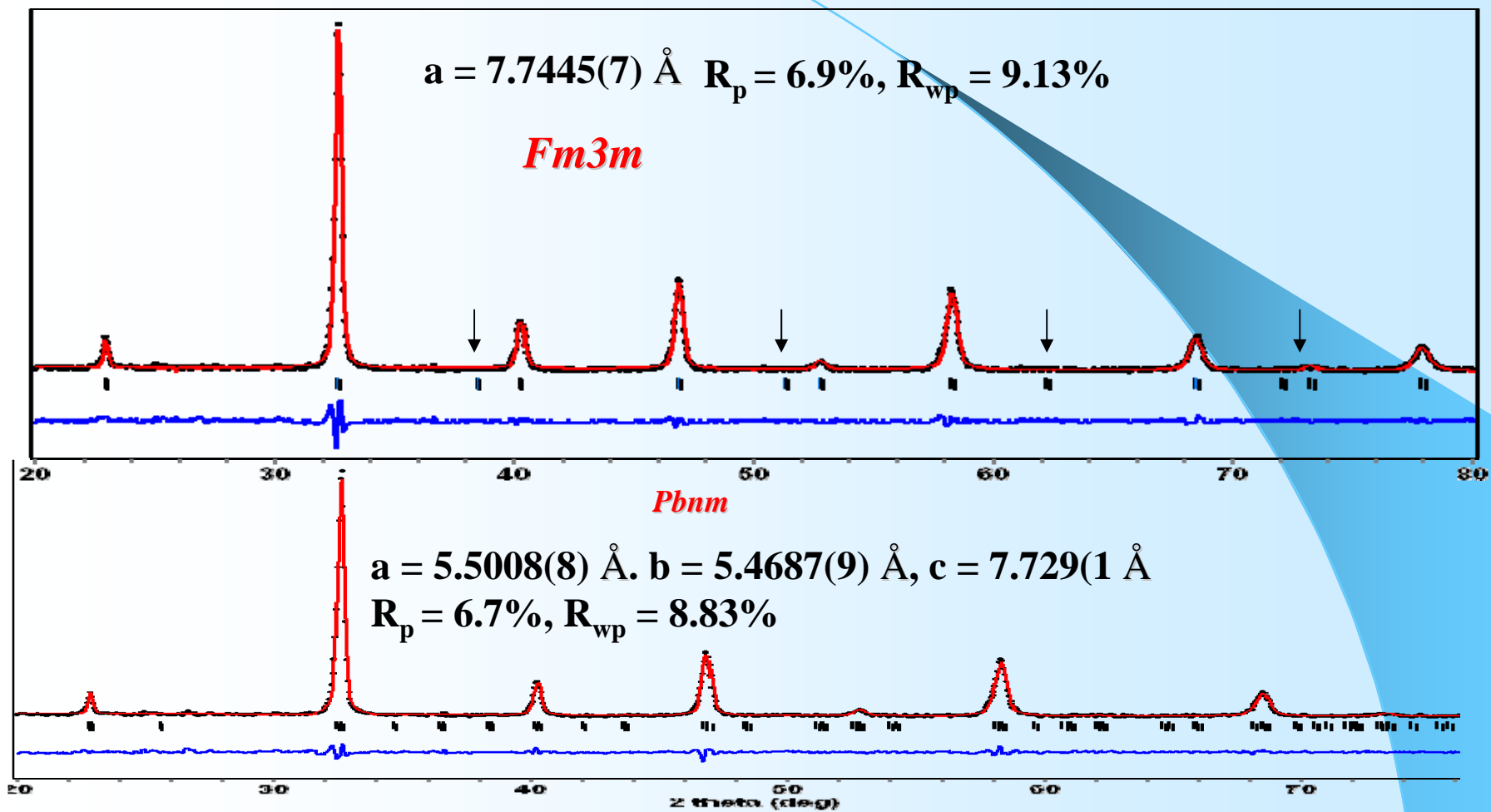


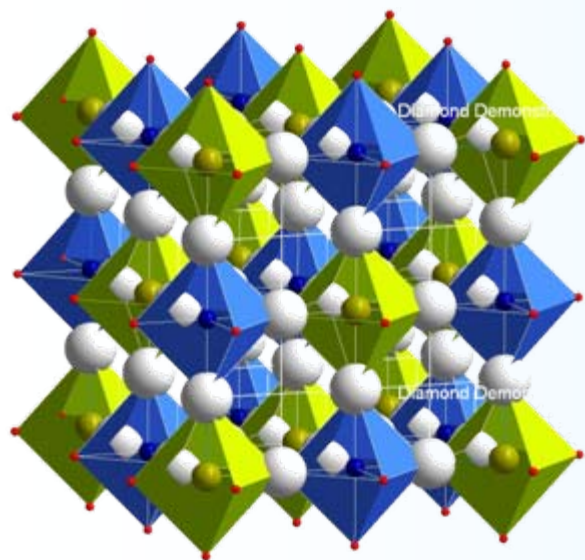
Fig. The X-ray diffraction pattern can be fitted in space group P2<sub>1</sub>/n (top) and Pbnm(bottom). Since the R-factors are comparable, it is not possible from laboratory X-ray diffraction measurements to assign the correct space group.

# SrLaMnFeO<sub>6</sub>

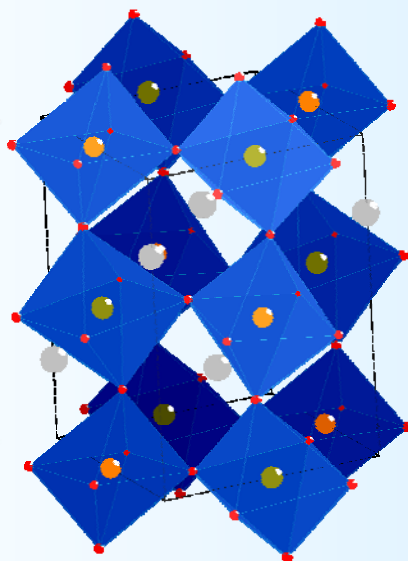


Refinement of X-ray diffraction data of SrLaMnFeO<sub>6</sub> in space group *Fm3m* (Top) and *Pbnm* (bottom). Arrows indicate unindexed reflections.

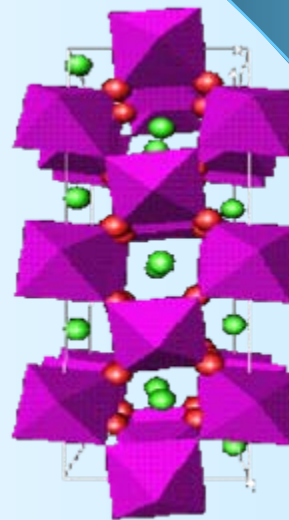
# What is the Crystal Structure of $\text{AlMnFeO}_6$ ?



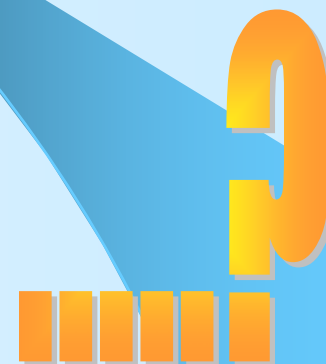
Cubic  
 $\text{Fm}\bar{3}\text{m}$



Monoclinic  
 $\text{P}2_1/\text{n}$



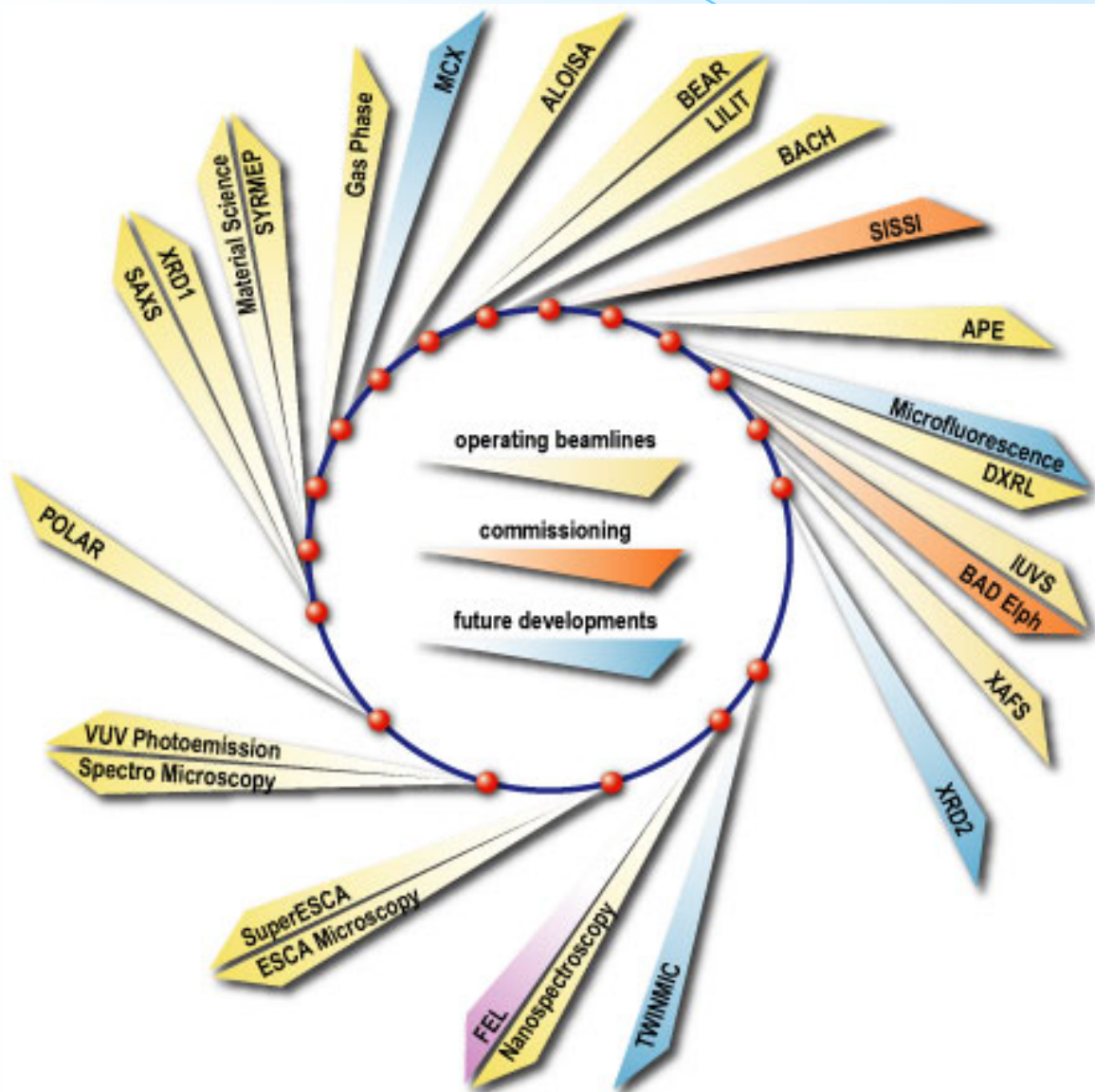
Orthorhombic  
 $\text{Pbnm}$







# ICTP-ELETTRA Users Programme



# ICTP-ELETTRA Users Programme

- The ICTP-ELETTRA Users Programme is offering access to the synchrotron radiation facility ELETTRA in Trieste in the years 2002-2006 to scientists who are citizens of developing countries and work in those countries. Up to an annual total of 1500 hours can be made available within this programme for beamtime applications at any of the existing ELETTRA beamlines.
- The programme is offering a limited number of grants to cover travel and living expenses of individuals and small groups who are meant to participate in the beamtime at ELETTRA. The number of scientists who can receive support depends on the number of allocated shifts and available funds:

# ICTP-ELETTRA Users Programme

In order to participate in the ICTP-ELETTRA Users Programme it is necessary to:

Submit an application for beam time following the usual ELETTRA procedure;

There are two deadlines every year:

- **February 28<sup>th</sup>**: for proposals eligible for the user period starting from July 1<sup>st</sup> to December 31<sup>st</sup>;
- **August 31<sup>st</sup>**: for proposals eligible for the user period starting from January 1<sup>st</sup> to June 30<sup>th</sup>.

The proposed experiments will be selected for beam time assignment on the basis of their scientific merit..